# Studying Properties of the Scintillation Spectra for some Solid Solutions under the X-Rays Excitation

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## $\Box$ ABSTRACT $\Box$

Scintillation characteristics such as spectra, decay times and yields of Cu- and Agdoped  $ZnS_XSe_{I-X}$  solid solutions, and some other compounds have been measured under Xrays excitation. Certain investigated materials show a high light yield in the long wavelength region and adequate kinetic properties. It is shown that the light yield can be increased for compounds synthesized at optimal conditions.

The measurements have been held in St. Petersburg State Polytechnical University, Petersburg, Russia

Key Words: Luminescent characteristic, Emission bands.

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دراسة مميزات أطياف التألق لبعض المحاليل الصلبة المتألقة تحت تأثير الأشعة السينية

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# 🗆 الملخّص 🗆

تحت تأثير الأشعة السينية، تم قياس مميزات أطياف التألق وأزمنة التفكك لعناصر النحاس والفضة المضافة إلى المحاليل الصلبة ZnS<sub>x</sub> Se<sub>1-x</sub> وكذلك بعض المركبات الأخرى المستخدمة.

أظهرت الدراسة على بعض المواد حدوث (تحرر) تألق ضمن مجال طول موجي واسع وبميزات طاقية كافية. وتبين أن طاقة الضوء المتحرر يمكن زيادتها في بعض المركبات المصنعة (المحضرة) عند شروط تركيبية مثالية.

أجريت القياسات في جامعة بطرس بورغ التكنولوجية الحكومية – بطرس بورغ – روسيا.

كلمات مفتاحية: مميزات التألق – الحزم الانبعاثية.

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#### Introduction

Commonly used silicon photodiodes exhibit an efficient sensitivity in the range from 500 to 1000 nm. This generates a need for long wavelength scintillators (LWS) [1]. There is a rich variety of phosphors and laser materials suitable for use in the red or near-infrared range, while most of the conventional scintillators emit in the short wavelength region. Generally, the long wavelength scintillators (LWS) can show high yield; and small forbidden bandwidth of a crystal is required for long wavelength emission. The Number of electron-hole (e-h) pairs generated by incident particle or quantum is inversely proportional to the band width of the crystal [2]. A large number of e-h pairs would yield high scintillation efficiency. A wide range of potential scintillation materials has been studied by Moses et al [3]. The crystals containing other rare-earth ions show longer decay times. So, the problem is to find materials with a short decay time among the long wavelength phosphors.

The properties of currently used LWS<sub>S</sub>, such as  $CsI:Tl,CdWO_4,Gd_2O_{2S}:Pr,Ce,$  $(YGd)_2O_3:Eu$  are a matter of common knowledge [4]. Some new compounds have been proposed as LWS<sub>S</sub>  $(\lambda m):Y_2W_2O_{12}:Eu$  (610*nm*),  $La_2O_2S:Eu$  (620*nm*) [5-7]

### Aim and importance of this work

In the present work we study the luminescent characteristics of  $ZnS_x Se_{l-x}$ (x = 0.5) solid solutions doped with Cu or Ag as activators and Al as co-activator and prepared as large transparent polycrystalline pieces. The Cu-doped samples exhibit broad emission bands, peaking near 600 nm. The best scintillation properties have been detected for ZnS - ZnSe : Cu, Al solid solutions.

#### **Methods and Experiments**

The ZnS - ZnSe solid solutions were obtained by means of sublimation in the form of polycrystalline unitary bodies. The synthesis was carried out by the physical vapor deposition (PVD) method in a closed reactor placed in a two-zone vertical furnace. Initially, the ZnS - ZnSe powder mixture was sintered at ~1000°C. Different Cu, Ag, and Al dopants and  $BaCl_2$ , CsCl, and NaCl fluxes were added at this stage of the synthesis process. The Al content in the samples after PVD was about 0.01%. 35mm diameter and 5 mm thickness samples were obtained. After that the samples were cut and polished.

The emission spectra of the samples were measured under steady-state X-ray (35V, 15mA) excitation, fig 1. A short pulsed (~1 ns) X-ray source and photon counting method were used for decay time measurements [8]. The measurements were carried out essentially at room temperature and in case of need in a 80-500 k temperature range. The energy efficiency ( $\eta$ ) of the scintillators was determined at pulsed X-ray excitation at 50 $\mu$ s shaping time. The ( $\eta$ ) value of the studied compound was compared with that for *CsI* :*Tl* crystal using it as a standard. The studied sample and *CsI* :*Tl* were placed as far as possible in similar conditions.

### **Results and Discussion**

In order to control the scintillation characteristics of the  $ZnS_x Se_{l-x}$  samples, we varied the value of x, deposition temperature, concentration of the dopants (Cu), and type and concentration of the fluxes (Table 1).

Both these features can give high scintillation efficiency [2]. The band gaps  $E_g$  are 3.6 and 2.67 eV in ZnS and ZnSe, respectively. Since the gap of  $ZnS_xSe_{l-x}$  is proportional to x [9], we can estimate that  $E_g \approx 3.14 eV$  for ZnS (0.5)-ZnSe (0.5) solid solution. Using similar considerations, we can assume that the density is ZnS (0.5)-ZnSe (0.5). The samples were reasonably transparent to self-emission. Absorption coefficient of ZnS-ZnSe: Cu, Al sample was  $\approx 7.0 cm^{-1}$  near 600 nm.

Table 1 Characteristics of some  $ZnS_X Se_{l-X}$  mixed samples: maximum of emission band  $\lambda_m$ , energy efficiency  $\eta$  and decay time constant  $\tau_1$  (X-ray excitation, room temperature).

No	X	Dopant (wt %)	Flux	$\lambda_m$ (nm)	$\eta$ of (CsI:Tl)	$\tau_1$ (ns)
1	0.5	Cu $(5 \times 10^{-3})$ , Al	CsCl	580	1.20	25
2	0.5	Cu $(5 \times 10^{-3})$ , Al	BaCl <sub>2</sub>	604	0.38	200
3	0.5	Cu (5 × 10 <sup>-2</sup> ), Al	NaCl	601	0.04	~1000

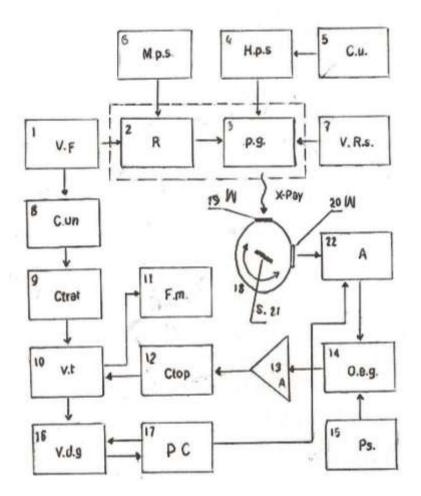


Fig 1: shows the instruments used for the measurement of the RL (X-Ray Luminescence), TSL (Thermoluminescence)

1- Variable frequency generator.	2- Rheostat.	
3- X-Ray pulses generator.	4- High power supply	
5- control unit.	6- Modulating power supply	
7-Variable power supply.	8- Cut universal.	
9-Time control connected to channel CYRAT.	10- Variable time modulator.	
11- Frequency measuring.	12-Time control connected to channel CTOP.	
13- Amplifier.	14- Optical electron generator.	
15- Power supply for optical electron generator.	16- Variable digital analyzers.	
17- IBM PC computer.	18- Cryostat.	
19- Windows for cryostat of Beryllium.	20- Quarts windows.	
21-Sample.	22- Analyzer.	

The X-ray excited emission spectra of some samples are shown in Fig 2. The growing condition (flux) does not influence strongly the emission shape of the ZnS (0.5)-ZnSe (0.5): Cu, Al Samples, while the intensity changes greatly. That is why the spectra of Fig 2 are presented on a semi-logarithmic scale. The samples exhibit broad emission bands peaking near 600 nm (Fig 2, Table 1). This is not particularly surprising, since the ZnS: Cu films [10] as well as ZnSe: Cu crystal [11] offer similar emissions. Increasing the ZnS content in the solid solution, the emission band shifts to the short wavelength region (Table 1). Related emission has been associated with rediative recombination of a free electron with a hole located in acceptor Cu<sup>+</sup> [12] or with a complex associative center [13]. Aluminum can play the role of a donor incorporated into associative luminescence centers [14]. An optimal content of Cu in in ZnS-ZnSe solid solution is about  $5 \times 10^{-3} wt \%$ . An increase of the Cu content leads to concentration quenching of the luminescence (curve 3, Fig 2).

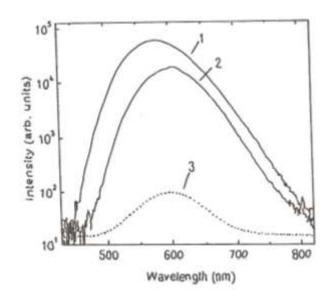


Fig 2: X-ray induced emission spectra of ZnS (0.5) ZnSe (0.5): Cu, Al compounds at room temperature (the curve number corresponds to the sample number in Table 1).

The scintillation efficiencies of the samples are presented in Table 1. The largest energy efficiency (120% of that of CsI:Tl) has been detected for ZnS (0.5)-ZnSe (0.5): Cu, Al growing with CsCl flux (Sample 1). The luminescence intensity of sample 1 (steady-state excitation) is about three times larger at (80K) compared with that at room temperature. Besides, a new emission band peaking at 525 nm appears at <140K.

Typical luminescence decay curves of some samples are presented in Fig. 3. The decay curves of the studied samples have usually complicated shapes. The initial parts of the curves may be approximated by exponents with decay constants  $\tau_1$ , which are presented in Table 1. For sample 1 (ZnS-ZnSe: Cu, Al), one can make out a fast component  $\tau_1 = 25$  ns and a slow component  $\tau_2 = 1.7 \ \mu$ s (curve 1, Fig 3). The last one has a high intensity and contributes significantly to the light yield of the sample. By contrast, sample 2 (BaCl<sub>2</sub> flux) exhibits one decay constant  $\tau_1 = 200$  ns and the emission decays to noise level in about 10  $\mu$ s (curve 2, Fig. 3).

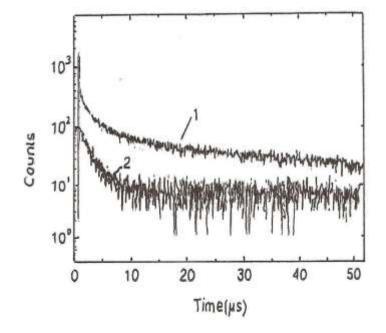


Fig. 3: Scintillation pulse shapes of X-ray excited samples: ZnS (0.5)-ZnSe (0.5): Cu (0.005), Al/CsCl (sample 1) -1 ZnS (0.5)-ZnSe (0.5): Cu (0.01), Al/NaCl (sample 3) -2

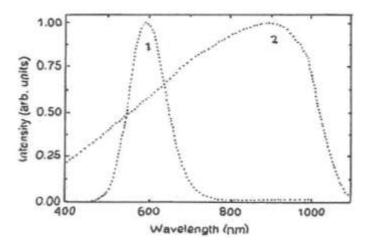


Fig. 4: X-ray excited emission spectra of ZnS/ZnSe: Cu, Al (1), and spectrum sensitivity of Si photodiode (2).

Compound	$\lambda_{\max}(nm)$	$\tau_1/\tau_2~(\mu s)$	$\eta$ of (CsI: Tl)
CsI: Tl	540	0.8	1
ZnS/ZnSe: Cu, Al	580-604	0.025/1.7	1.20
LaMgAl <sub>11</sub> O <sub>19</sub> : Nd, Cr	695, 709, 750	0.45	-
$Sr_2P_2O_7$ : Pr	520, 594, 603, 640	0.3	-
$CdI_2$	505, 680	0.004	0.20
CaMoO <sub>4</sub>	532	~0.004/16	0.20

Table 2 Main scintillation characteristics of some compounds

Several others compounds have been investigated in assumption that they give long wavelength scintillations.

Main characteristics of the compounds are presented in Table 2 and spectra of some of them are shown in Fig.5. Unfortunately,  $CaF_2$ : Na and  $SrF_2$ : Na crystals, which show an efficient long wavelength luminescence of  $F_{2A}$  centers at optical (UV) excitation, do not scintillate at X-ray excitation. It means that the energy transfer from host to  $F_{2A}$  centers in the crystals is not efficient.

#### Conclusions

The present work shows that the characteristics of some studied scintillators can be improved. Therefore, we have to choose optimal growth conditions, concentration of activator and co-activator, and the type of flux. Besides, we plan to vary the post-growth conditions, particularly the cooling rate.

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